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Fabrication of Three Terminal Devices by ElectroSpray Deposition of Graphene Nanoribbons

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Abstract

Electrospray deposition (ESD) in ambient conditions has been used to deposit graphene nanoribbons (GNRs) dispersed in liquid phase onto different types substrates, including ones suitable for electrical transport. The deposition process was controlled and optimized by using Raman spectroscopy, Atomic Force Microscopy and Scanning Electron Microscopy. When deposited on graphitic electrodes, GNRs were used as semi-conducting channel in three terminal devices showing gate tunability of the electrical current. These results suggest that ESD technique can be used as an effective tool to deposit chemically synthesized GNRs onto substrates of interest for technological applications.

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1.Introduction

The use of graphene in electronics is attracting much interest for its potential as constituent material of several devices like transistors, sensors, touchscreens and energy storage devices [1]. Its excellent charge mobility and transparency to the visible light are well established and ready to be exploited, but the use of graphene in electronic devices, such as a field effect transistor (FET), requires the presence of an energy gap in the electronic band structure and viable strategies to achieve this goal are still under exploration. The reduction of the lateral size of the graphene layers down to graphene nanoribbons (GNRs) is an efficient way to induce an energy gap, however as long as the GNRs are obtained by conventional top-down approaches such as lithography [2] or by unzipping of carbon nanotubes [3], the resulting structure is not controllable, making the electronic properties unpredictable. The use of GNRs with atomically-controlled size is therefore very appealing in this context. Typically, high-quality GNRs are grown in ultra high vacuum on metallic surfaces [4,5], however non-conducting substrate is mandatory for applications in electronics. GNRs with atomically defined edges can be alternatively synthesized in liquid-phase [6,7]; given their huge mass, they cannot be evaporated and the deposition is usually performed by drop casting, possibly functionalizing the surface of insulating substrates [8-10].

Electrospray deposition (ESD) [11-15] is an alternative technique that allows to softly land large and heavy molecules from liquid suspension on to any substrate. ESD has been successfully used for the deposition of giant biomolecules (proteins) [16], fullerenes [17,18], carbon nanotubes [19], molecular nanomagnets [20-23] and nanoparticles [24], thus enabling integration of large molecular units in electronic circuits. Homemade and commercial apparatuses are proposed for ESD in a large variety of designs and sophistications. Briefly, besides the capillary injecting the desired sprayed solution/suspension and the target holding the desired support, an ESD setup may include: 1) stages under differential pressure, leading to an UHV environment where the support (either conducting or insulator) is maintained; 2) one or more stages of electromagnetic lenses that act as mass selectors and/or to (de-)accelerate molecules.

Here we report a feasibility study for the fabrication of three terminal electronic devices made of chemically synthesized GNRs dispersed in tetrahydrofuran (THF) and

deposited by an elementary ESD setup. The idea is to use *jugaad* approach to test the viability of this route. To this end, we control the morphology of the deposited GNR film by optical microscopy, scanning electron microscope (SEM), atomic force microscopy (AFM) and the quantity of deposited material by semi-quantitative Raman spectroscopy. To prove the effectiveness of our approach, GNRs have been deposited on graphitic electrodes and I - V characteristics measured at room temperature, showing a pronounced gate tunability of the device current.

2. Experimental Methods

2.1 Electro spray apparatus

The ESD system used in our experiments is the early stage of a *Thermo Finnigan Surveyor* Mass Spectrometer (MSQ). The quadrupole and the mass analyzer stage were not used due to the huge mass of the GNRs having a molecular weight in the order of 10^4 – 10^5 u. Briefly (see Fig. 1), our Electrospray source is composed by a stainless steel insert capillary, where the solution flows. The capillary is hold at voltage V_c , tunable in a range of ± 5 kV, with respect to the target and the rest of the chamber. It is surrounded by a tube injecting nitrogen and the action of both the gas and the voltage V_c produces an aerosol of liquid droplets. The nitrogen promotes the solvent evaporation from the flying charged droplets: when their charge exceeds the surface tension, fission occurs producing small droplets [25]. The pressure of the nitrogen gas was kept at 4 bar at the entrance of the tube in order to get high flux of clean nitrogen in the whole deposition chamber. The solution is contained in an external syringe and it is driven into the capillary by a pump.

The temperature of the capillary and its casing is tunable up to 650°C. High temperature facilitates the evaporation of the solvent and impedes the blocking of the capillary due to the accumulation of agglomerates. The quantity of solute that is deposited was controlled by the time of surface exposure to the flow and indirectly by the concentration of the solution, while the rate of the syringe pump was kept constant. All the depositions have been carried out in ambient conditions, namely at room temperature and atmospheric pressure.

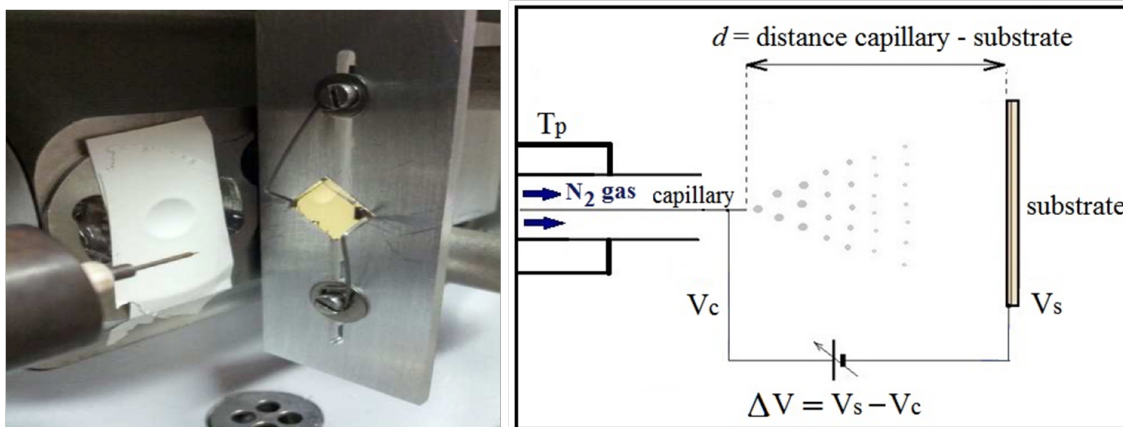


Figure 1: Scheme of the ESD system with the relevant parameters: the probe temperature (T_p), the capillary voltage (V_c) and the sample voltage (V_s) and the distance between substrate and capillary (d).

2.2 Graphene nanoribbons and their dispersion

In this work we used structurally well-defined GNRs with a “cove”-type edge structure and a width of approximately 1 nm (Fig. 2), which were bottom-up synthesized by the methods of synthetic organic chemistry, following a reported procedure [6]. Briefly, the synthesis was carried out through the AB-type Diels-Alder polymerization of a monomeric precursor 2,5-bis-(4-dodecylphenyl)-3-(3-ethynylphenyl)-4-phenyl-2,4-cyclopentadienone to provide non-planar polyphenylene precursors, which were subsequently “graphitized” by oxidative cyclodehydrogenation in solution. Uniform GNRs with the same width and edge structure have thus obtained as powder samples, which could be dispersed in organic solvents such as THF, thanks to the fact that they possess dodecyl ($C_{12}H_{25}$) chains attached at the peripheral positions. Relatively short GNR samples (<100 nm) have been used for the following ESD experiments, considering the decreasing dispersibility of longer GNRs. Two GNR samples with different estimated average lengths, namely approximately 20 and 80 nm, have been examined and gave identical results, at least upon our analysis. Here, the average lengths were roughly estimated based on molecular modeling and weight-average

molecular weights of the corresponding polyphenylene precursors, obtained by the gel permeation chromatography analyses against polystyrene standards.

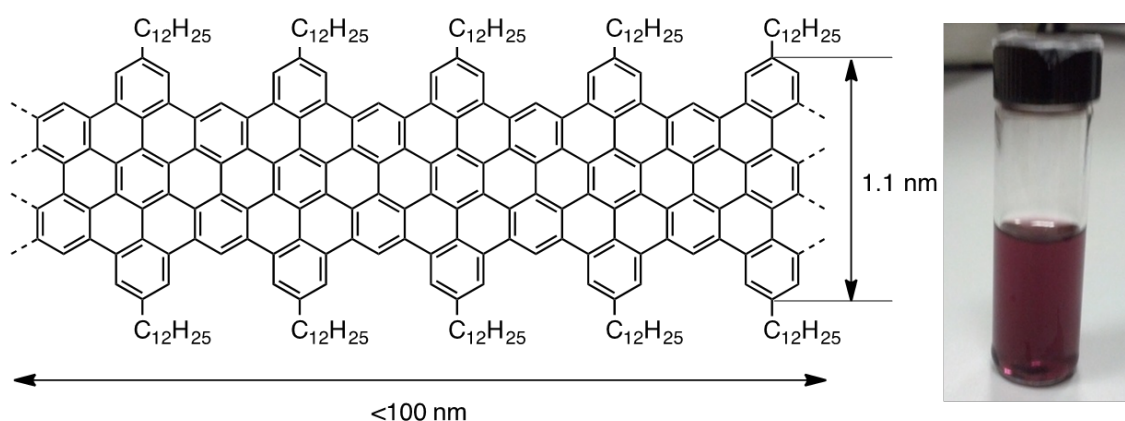


Figure 2: Left: Chemical structure of the GNRs used for the depositions. Right: Mother dispersion S_0 of the GNRs in THF.

In the following, we define "mother dispersion" (S_0) as a supersaturated mixture of GNRs in THF. To obtain good diluted dispersion, S_0 has to be deeply sonicated. To change the concentration, we draw part of S_0 far from the precipitate by a pipette and add THF to make an intermediate dispersion S_1 . We refer to $S_1=10$ (100) to a mixture prepared by diluting S_0 10 (100) times with THF. Since THF is slightly corrosive for the ESD components, the dispersion S_1 is further diluted 10 times with acetonitrile (CH_3CN) to obtain the dispersion S_2 that is actually used for ESD. Since the latter step is performed for all the samples, the concentration of the deposited material is therefore defined only by the dispersion S_1 . After the dilution, S_2 was sonicated once more for at least 15 minutes at 40 °C in order to reduce the formation of GNR aggregates. In early tests we have also used methanol instead of acetonitrile, but in this case it was quite difficult to maintain good dispersion and the deposition needed to be done immediately after sonication. The quantity of deposited material can be controlled by changing the concentration of the dispersion S_1 and the time of deposition since the rate of syringe pumping is constant and computer controlled, thus in the following we report the quantity of

deposited material in *ml* of sprayed S₂ dispersion. In our experiments the time of deposition ranged between 10 sec to 40 minutes, suggesting that this approach is a rather fast and efficient way to prepare films of GNRs on different types of substrates.

2.3 Surface techniques

The morphology has been firstly visualized by means of optical microscope (Olympus-BX51M), SEM, AFM and scanning tunneling microscopy (STM), reported in Supplementary Information, at different scales.

We used a Veeco Multi Mode Nano Scope IIIa AFM tuned in tapping mode.

The SEM images were accomplished through the Carl Zeiss Sigma equipped with the GEMINI column (typical acceleration voltage used=2kV) while the STM measurements were carried out by an Omicron UHV VT STM system.

Raman spectra have been collected by using a Jobin Yvorn LabRAM with an integrated optical microscope, through which we can select the area to be illuminated. We used a laser's wavelength $\lambda = 632.81$ nm and the diameter of the spot was ≈ 1 μm in our typical configuration.

3. Results

3.1 Investigation by optical microscope

We have initially optimized the deposition parameters by systematically changing concentrations and modifying the distance d between the capillary and the substrate (see Fig. 1). Quantitative information was obtained by Raman analysis of the characteristic D and G peaks.

Firstly, we used the optical microscope for a preliminary investigation on the deposition.

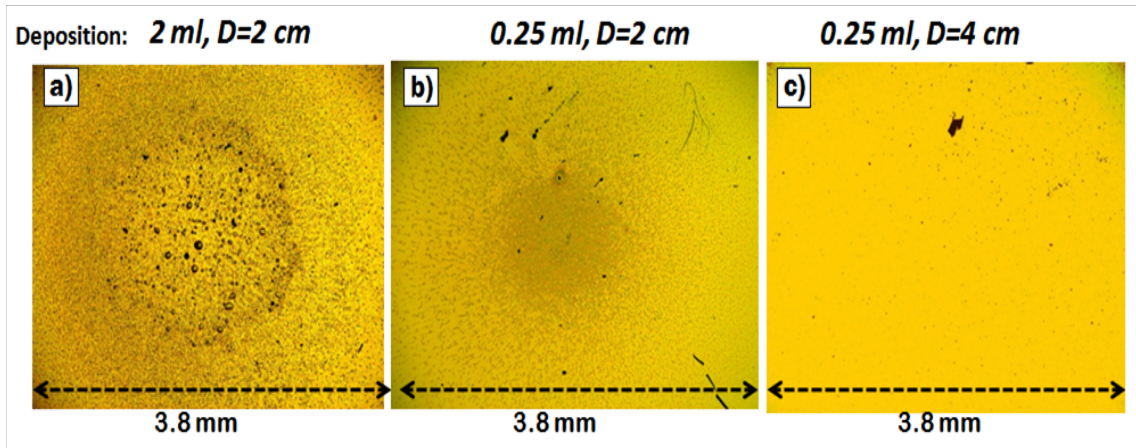


Figure 3: Magnification $5\times$ optical images of Au/mica after ESD of the GNRs with different quantity of dispersion ($S_1=10$) and different distance d between capillary and substrate: a) 2 ml deposited at $d=2$ cm. b) 0.25 ml deposited at $d=2$ cm. c) 0.25 ml deposited at $d=4$ cm.

When the substrate is close to the capillary ($d=2$ cm), a central halo is visible in Figure 3a with the clear formation of large aggregates of GNRs. By decreasing the deposited quantity, the homogeneity of the halo increases and the formation of clusters reduces (Fig. 3b). Since the flux injected by the capillary has a conical shape, larger distance d between the capillary and the substrate increases the deposition area and the film homogeneity, as a consequence of the reduction of the droplets density. Thus we increased the distance between the substrate and capillary to $d=4$ cm (Fig. 3c), obtaining a more homogeneous film where the central spot is no longer visible at optical microscope. This test suggests that it is possible to efficiently cover large areas ($\sim\text{cm}^2$) by this technique.

3.2 Investigation by Raman Spectroscopy.

Raman spectroscopy is an efficient tool for studying carbon-based materials [26]. In particular the G peak at $\sim 1600\text{cm}^{-1}$ (which is induced by the in-plane optical phonon branch and therefore is found in all carbon allotropes with sp^2 hybridization) and the D peak at $\sim 1300\text{cm}^{-1}$ (which is a second order process, in which out-of-plane optical phonons scatter with a defects) are well established in literature to characterize the GNRs [27-30] and a careful analysis of their shape would allow to get information also about the GNR edge [28]. We have simply performed a comparative study of these

peaks to get semi-quantitative information on the quantity of deposited carbon material. A similar approach has been previously employed to study the deposition of carbon nanotubes and the growth of hydrogenated diamond-like carbon films prepared by plasma deposition [31,32]. The background signal due to residual solvent or amorphous material was estimated and subtracted before the measurements of the GNR-deposited samples. In general, the intensity of a Raman peak depends on the laser power, spot size and the penetration depth of the beam. We have therefore took care to perform all the measurement in the same conditions and in a short period of time to avoid the deviation of the laser power. Statistics over the deposited surface was also necessary to account for thickness inhomogeneity (see Supplementary Information for more details).

Fig. 4b and 4c show the trend of the G and D peaks intensity for different deposition conditions on Au/mica. The intensity of both the Raman peaks clearly scales with the quantity of deposited GNR as estimated by the deposition parameters. Similar trends are also obtained considering the area of the respective peaks (see Fig. S2 in Supplementary Information). By decreasing the quantity of deposited GNRs, intensity and areas of both G and D peaks get very small and close to the limit of sensitivity of our measurements that we roughly estimate to be close to 1 monolayer (ML) by comparison with other type of samples.

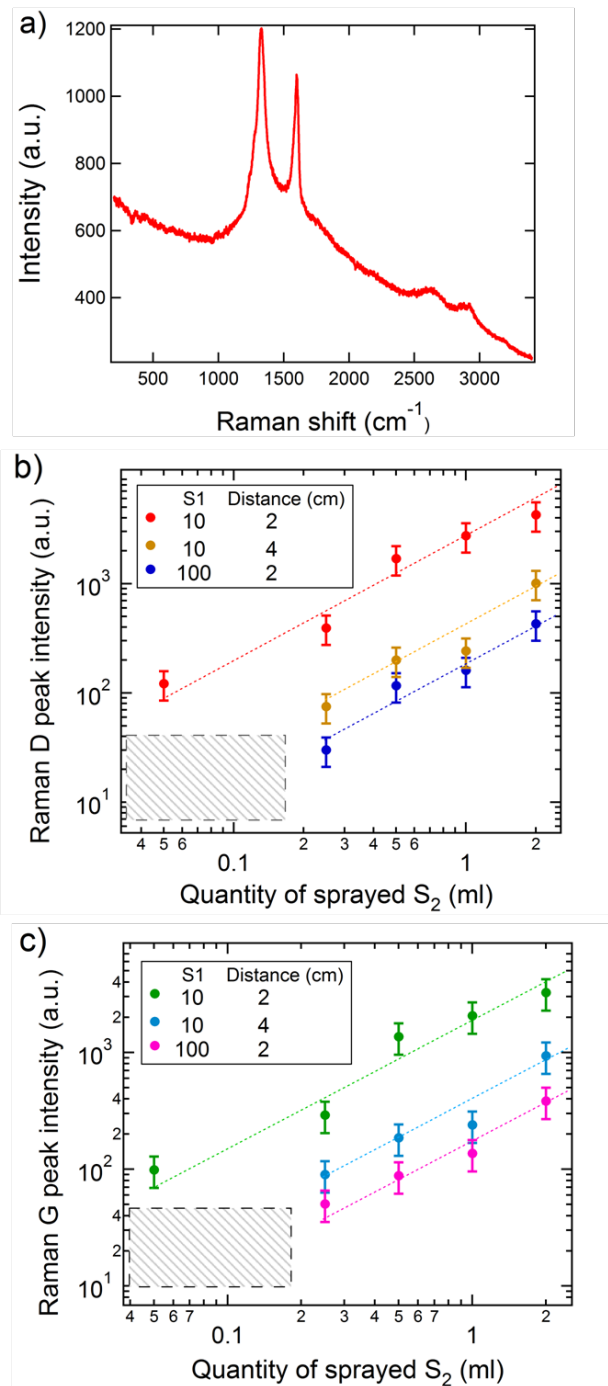


Figure 4: a) Typical Raman spectrum of a GNR film deposited by ESD on Au/mica. b) Intensity of the D peak as a function of the quantity of S₂ deposited for three different deposition conditions. c) Intensity of the G peak.

3.3 Investigation by Scanning Microscopies

To verify the morphology of the surface we used an AFM and a SEM.

Fig. 5 shows AFM images taken in the central region of a GNRs spot deposited on a Au/mica substrate. The deposition parameters were tuned to obtain one of the thinner GNR films according to the semi-quantitative information acquired from Raman (see Fig. 4). In Figure 5a, a network of GNRs bundles is visible at the spot center and we can recognize structures that can be associated to the GNRs with typical dimensions of 2-3 nm in height and 10 to \approx 100 nm in length (Fig. 5 b, c, d).

Similar results have been obtained by depositing GNRs also on the insulating SiO₂ substrate. Large substrate areas (200 μ m wide) are homogeneously covered by GNRs aggregates: an optical image of 0.25 ml of solution sprayed on a 300 nm thickness of SiO₂ surface over a Si bulk is shown in Fig. 6a. The formation of GNRs aggregates is confirmed by Raman spectra (Fig. 6b) whose intensity at the D and G peaks is very strong on purple spots. On the other hand, in the white shadows the Raman signal is much less intense (Fig. 6c) and it almost disappears (but still detectable) in the blue areas of the optical image. Both the SEM and AFM images better display the morphology of these aggregates, which is probably determined by the slow evaporation of residual solvent.

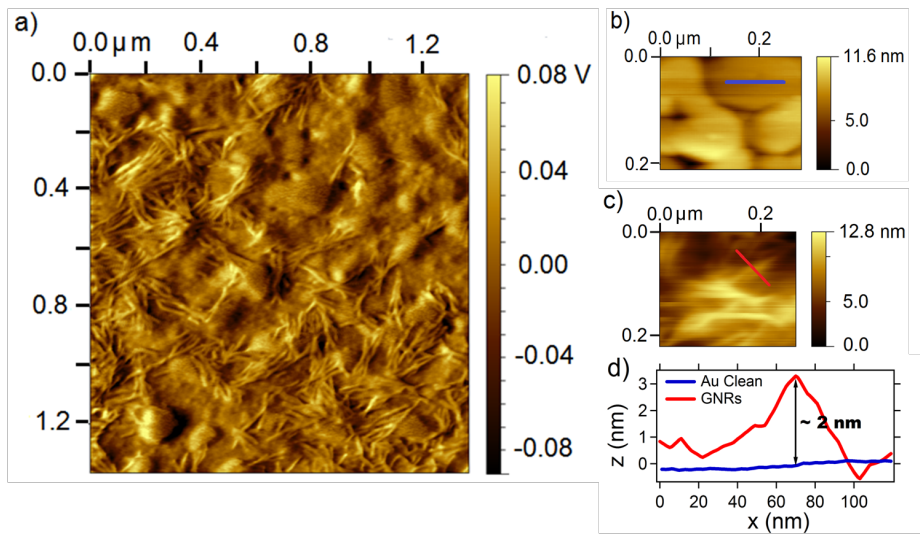


Figure 5: a) Amplitude image of GNRs deposited by ESD of 0.05 ml of S_2 (with $S_1=10$) on Au/mica, distance between capillary and Au/mica = 2 cm. b) AFM image of clean Au/mica substrate. c) ESD of 0.05 ml of S_2 (with $S_1=10$) on Au/mica. d) Line profiles shown in the images b) and c).

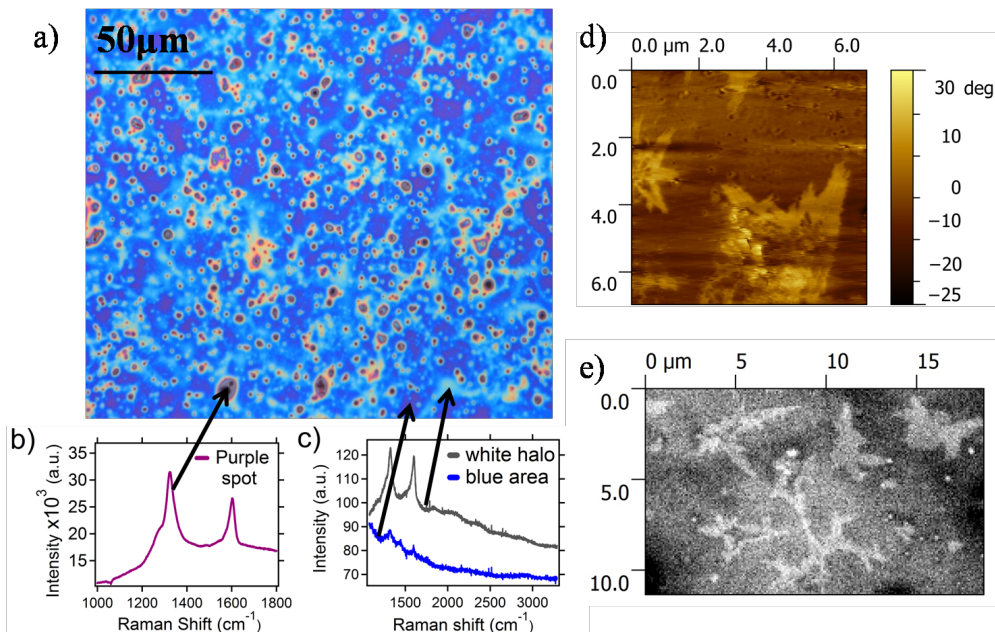


Figure 6: a) Magnification $100\times$ optical image of GNRs deposited by ESD on SiO_2 . 0.25 ml of $S_1=10$, $d=4$ cm. b) Typical Raman spectrum on the purple spots present on the substrate. c) Typical Raman spectrum on the white regions that surround the purple spots (grey spectrum) and on the blue areas (blue spectrum). d) AFM phase image of the substrate. e) SEM image of the substrate (Magnification = $1200\times$).

3.4 Fabrication and characterization of three terminal device

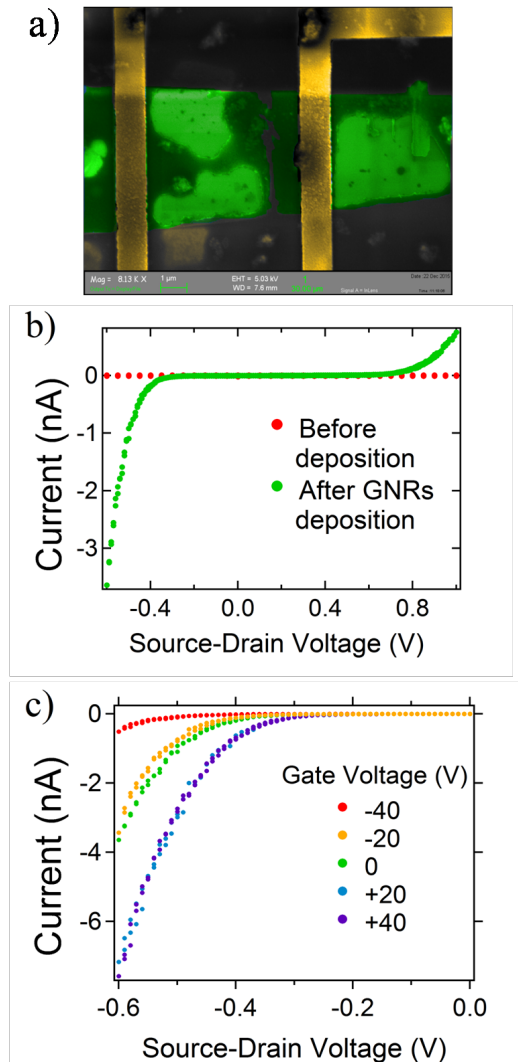


Figure 7: a) SEM image of graphitic contacts made by the electro-burning process [33] on which we deposited 1 ml of S_2 (with $S_1=10$). The yellow stripes are Cr/Au contacts patterned by EBL, the darker area in the graphene flake (green) is material electro-sprayed on the device: it covers a part of the flake and in particular the gap done by electro-burning. b) I-V characteristics of an electro-burned device before and after the deposition of GNRs through ESD. c) I-V characteristics for different values of the gate voltage.

To test the effectiveness of the ESD method for the realization of GNR-based electronic devices, we have deposited the GNRs on graphitic electrodes on top of a 300-nm-thick SiO₂ substrate, using the underlying doped silicon to make a back gate. The graphitic electrodes were prepared using the electro-burning process [33]. Briefly, multilayer graphene flakes are mechanically exfoliated on the substrate and Cr/Au contacts (typically spaced 1 to 6 μm one to another, see Fig.7a) are then fabricated by electron beam lithography (EBL). A physical gap is then opened in the graphitic channel by the electro-burning process [33], setting the parameters to obtain an aperture of the order of ~10-50 nm. The opening of the gap can be recognized also by the *I-V* characteristics taken after the electro-burning process, finding no measurable current in the range of –2V to +2V (Fig.7b). After confirming the successful fabrication of the graphitic electrodes (no current passing between source and drain), we deposit GNRs by ESD on the whole substrate and measure again the conductance of the devices. The quantity of sprayed solution ($S_2 = 1\text{ml}$) makes reasonable the probability to get GNRs in the region of the gap (Fig. 7a): on a chip with initially ~10 electro-burned devices, two of them show a significant increase of conductivity after the GNR deposition (Fig. 7b).

To get reproducible *I-V* characteristics, an annealing at 130°C in vacuum (10^{-7} mbar) is necessary: this likely makes the GNR film more stable on the surface, improving the electrical contact with the graphitic electrodes. After this annealing, the source-drain current I_{SD} is in the nA range for $V_{SD} = 0.5\text{V}$. Non linear *I-V* curve probably reflects Schottky barrier at the interface with the electrodes. Interestingly, such *I-V* characteristics show a pronounced gate dependence (Figure 7c), with a current on/off ratio as high as 5. This is higher than what reported for this type of solution-synthesized GNRs [8,9], proving the effectiveness of the ESD as fabrication method. In the previous reports on electrical devices with this type of GNRs the poor gate tunability was generally ascribed to the tendency of the GNRs to form huge aggregates, as the electric field from the back gate is screened by the adjacent ribbons [8,9]. In addition, it has been shown by DFT calculations that the graphene-like band structure (with a vanishingly small energy gap) can be restored as a consequence of the ribbon-ribbon interaction [9]. Here we believe that the electrostatic repulsion between the sprayed GNRs helps to reduce their aggregation, increasing the final response to the applied electric field.

4. Discussion and Conclusions

In this work we have used a simple setup to spray ultra-long solution-synthesized GNRs dispersed in organic solvents. This approach allowed us to circumvent intrinsic and technical difficulties (*e.g.* transfer of GNR on non conducting substrate, evaporation of large mass molecules etc.) that are limiting the use of GNR in electronics. The use of combined analysis allowed us to show that ESD is an efficient way to deposit GNRs on any type of substrates (we tested metallic Au/mica, non conducting SiO₂), and to cover large area as shown by optical microscopy, SEM and AFM on different scales (μm^2 to cm^2). We used semi-quantitative analysis of the G and D Raman peaks to quantify the amount of deposited material and show that it is possible to control the quantity of material deposited through the feeding electrospray source. Three terminal electronic devices have been successfully fabricated by electrospray GNRs onto a SiO₂ substrate with graphitic electrodes. I - V characteristics show finite current (nA at $V_{\text{SD}}=0.5\text{V}$) through the GNR channel and dependence to the gate voltage. Overall, these results show that ESD is a fast deposition technique that represents a valid alternative to drop casting for deposition in ambient conditions.

Further improvements are readily visible adopting suitable technical solutions. For instance, GNRs film homogeneity can be further improved by placing the substrate more distant from the spraying capillary. Additional improvement can potentially be obtained by depositing in a chamber with differential pressure: this may facilitate formation of smaller droplets and improve homogeneity over large surfaces. In commercial mass spectrometer such chamber is placed off axis and after selection of charged molecules and, in the case of GNR, its use results in a drastic reduction of deposition rates. This can be suitable for fabrication of devices comprising single GNR. Alternatively different designs with substrate facing the capillary after chambers with differential pressure may allow efficient deposition also in UHV conditions.

Another interesting parameter of ESD is the application of high potential $\Delta V=V_c-V_s$ between the capillary and the substrates. This has essentially two roles: firstly it ionizes the droplets thus facilitating their dispersion; secondly it charges GNRs. The latter process is necessary to accelerate the GNRs and to select their masses if spectrometer is used. Yet, because of the large mass of GNRs, this process is scarcely

effective and it can be exploited only with a dedicated apparatus. Nevertheless, charged GNRs tend to repulse each other and to better stick on surface. Thus, even if electromagnetic lenses are not used for mass selection, the use of charging potential can help in depositing isolated GNRs on substrate and this can be a further parameter to be controlled.

Acknowledgements.

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